

## The Structure of a Barium Niobium Silicon Oxide with the Probable Composition $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ ( $x \approx 0.23$ )

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Reaction of  $\text{BaO}$ ,  $\text{Nb}_2\text{O}_5$ , and  $\text{Nb}$  in mole ratios of 2.4:1.6:1 in an evacuated silica capsule at  $1250^\circ\text{C}$  produces a mixture of at least two products, one of which has the probable composition  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$  ( $x \approx 0.23$ ). This compound has a hexagonal unit cell of dimensions  $a = 9.034 \pm 0.004 \text{ \AA}$ ,  $c = 27.81 \pm 0.02 \text{ \AA}$ , probable space group  $P6_3/mcm$ ,  $Z = 2$ . Its structure has been determined from 942 independent reflections collected by a counter technique and refined by least squares methods to a conventional  $R$  value of 0.062. The basic structure consists of strings of four  $\text{NbO}_6$  octahedra sharing opposite corners, each string joined to the next by edge sharing of the end octahedra, so that the  $c$  axis corresponds to the length of a strand of seven corner-linked octahedra. Chains of three such strands are formed by corner sharing between the strands. The chains in turn are joined by  $\text{NbO}_6$  octahedra and  $\text{Si}_2\text{O}_7$  groups in which the  $\text{Si-O-Si}$  linkage is linear. Barium atoms are in sites between the chains coordinated by 13 oxygen atoms. A second site, 15 coordinated, probably has a small amount of barium as well; the fractional occupancy for barium in this site is 0.076.

### Introduction

The reaction of  $\text{BaO}$ ,  $\text{Nb}_2\text{O}_5$ , and  $\text{Nb}$  in the presence of  $\text{SiO}_2$  can result in at least three different hexagonal complex oxides containing silicon. The first to be studied was  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$  (1). This structure has infinite strings of  $\text{NbO}_6$  octahedra sharing opposite corners. These strings extend in the  $c$  direction. Each string is joined laterally to two others by corner sharing to form chains; the chains in turn are linked by corner sharing with  $\text{Si}_2\text{O}_7$  groups and by the electrostatic attraction of  $\text{Ba}^{2+}$  ions between the chains. A second compound,  $\text{Ba}_3\text{Nb}_{21-x}\text{Si}_2\text{O}_{44}$  ( $x \approx 0.2$ ) has a different structure based on a 7 layer stacking of close-packed  $\text{O}_7$  and  $\text{BaO}_6$  layers with niobium and silicon occupying sites between the layers (2).

The compound described here,  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ , has a structure related to that of  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$ . However, it is more complex and has some points of individual interest.

### Experimental Methods

$\text{BaO}$  was prepared from  $\text{BaO}_2$  by heating to  $1000^\circ\text{C}$  *in vacuo* for 1 hr. It was mixed with  $\text{Nb}_2\text{O}_5$  and  $\text{Nb}$  (both obtained commercially) in a 2.4:1.6:1 mole ratio; the resulting gray powder was triturated under petroleum ether and placed in a silica capsule, which served as both container and reactant. The capsule was evacuated, sealed, and placed in a furnace at  $1250^\circ\text{C}$  for 1 wk. Upon removal from the furnace, the capsule was in a partially collapsed state, and on cooling it clouded and became crazed on the inside.

The reaction product consisted of black, highly reflective crystals, some embedded in the container walls, some aggregated into chunks that appeared to have been partially melted, and some loose and apparently single. The last-named occurred in two distinct habits: thick chunks only occasionally exhibiting flat faces, and very thin hexagonal plates, rarely complete but usually having at least one  $120^\circ$  angle visible.

Examination of several of the single crystals indicated that these two types of crystal represented two different compounds. The thin plates proved to be  $\text{Ba}_3\text{Nb}_{21-x}\text{Si}_2\text{O}_{44}$  and the chunks  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ . A rough density of  $4.9 \text{ g/cm}^3$  was measured for the aggregated crystals (by the method of Archimedes, using water as the displacement liquid), which presumably consist mainly of the latter compound. This value is probably low because of trapped air in the aggregates.

A single chunky crystal was selected for structure determination and ground to a sphere of 0.14 mm diameter. Laue and precession X-ray photographs revealed a hexagonal lattice, Laue group  $6/mmm$ , with  $hhl$  reflections present only for  $l = 2n$ . The probable space group is thus one of  $P6_3cm$ ,  $P6c2$ , and  $P6_3/mcm$ . Photographic measurements gave  $a = 9.00 \text{ \AA}$  and  $c = 27.67 \text{ \AA}$ .

Data for the structure solution were collected on a Picker Nuclear computer-controlled single-crystal diffractometer, using graphite-monochromatized Mo radiation and a scintillation counter detector, and operated in the  $\theta$ - $2\theta$  scan mode. The base scan width was  $2.5$ - $2.8^\circ$  in  $2\theta$ , with dispersion factor of 0.692. Background was counted for 20 sec on each side of the peak, and three standard reflections were measured every 40 reflections. In all, three sets of data were collected. Due to errors in interpreting the photographic data, the first collection was based on an assumed  $c$  value half the true one and hence missed half the reflections. The data produced were not used in the structure determination, although the final cell parameters,  $a = 9.034 \pm 0.004 \text{ \AA}$  and  $c = 27.81 \pm 0.02 \text{ \AA}$  ( $= 2 \times (13.906 \pm 0.009 \text{ \AA})$ ), were determined at this time by least squares refinement on the diffractometer angles of 12 reflections.

The second collection produced usable data, but instrument instabilities resulted in sizable variations in the standards with time. Dividing the data into a number of groups and scaling each group to bring standards into agreement gave standard deviations of the standards of 1.6, 2.6, and 5.1%. However, the data were still not of very good quality, and when the conventional  $R$  value would go no lower than 0.124 for the final refined structure, a third set of data was collected. It was later discovered that part of the refinement problem arose because we were attempting to refine a nearly centrosymmetric structure.

The third data set included all possible reflections with  $2\theta \leq 25^\circ$ , and those within the

range  $0 \leq h \leq 9$ ,  $-3 \leq k \leq 9$ , and  $-3 \leq l \leq 33$  for  $25^\circ < 2\theta \leq 50^\circ$ . The three standards had standard deviations of 1.4, 2.2, and 5.4%, with no scaling necessary. The 3329 reflections collected were corrected for background and those related by symmetry were averaged, reducing the data set to 670 reflections, of which 421 had intensities greater than three times the standard deviation of the background and were labeled observed. In the table of structure factors, the number listed in the column of observed  $F$ 's for unobserved reflections is determined from  $I_{\text{obs}} + 2\sigma(I_{\text{obs}})$ .

The data set used for the final structure determination included all the reflections from the third data collection plus 242 reflections of raw intensity greater than 100 and  $2\theta > 50^\circ$  from the second data collection. (The intensity restriction was imposed because of the pronounced inverse correlation between magnitude and  $R$  factor for the second-set reflections.) Lorentz and polarization corrections and a spherical absorption correction, with  $\mu r = 0.76$ , were applied by the X-ray system program package (3), which was also used in subsequent calculations of Fourier maps, least squares refinements, and so forth.

### Structure Determination and Refinement

The  $hk0$  reflections of the compound under study are very similar to those of  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$ , indicating a resemblance between the projections along  $c$  for the two compounds. Ba atoms were therefore placed at 0.59, 0,  $z$  positions; Nb at 0.24, 0,  $z$  positions and Si at  $1/3$ ,  $2/3$ ,  $z$  positions in the present compound to correspond to  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$ . Assuming a barium at 0.59, 0, 0, a Patterson map indicated possible niobium  $z$  values of 0.072, 0.211, and 0.350. The four-atom model thus arrived at was used as the starting point for a structure solution by difference Fourier methods.  $P6_3cm$  was chosen as the trial space group.

The niobium position and population parameters and the barium  $x$ -parameter were refined and a  $\Delta F$  map was calculated. This was then used to place further atoms and the cycle repeated, with isotropic thermal parameters also being refined on later iterations. After a few such cycles,  $R$  was down to the 10% range and the model was making chemical sense. At this point the indicated chemical formula was  $\text{Ba}_6\text{Nb}_{14}\text{Si}_4\text{O}_{50}$ , with  $Z = 2$ .

Anisotropic thermal parameters were introduced for barium and niobium and all possible parameters refined. This process reduced the  $R$  factor by a few percent, but the parameters would not converge. Moreover, the thermal parameters for one niobium and several oxygens became nonpositive definite. A weighting function  $w = 1/\sigma_F^2$  was introduced, with

$$\sigma_F = \frac{1}{(n)^{1/2}} \frac{1}{2(Lp)^{1/2}} \times \{[N_T + BG + [0.02(N_T - BG)]^2] / [N_T - BG]\}^{1/2}$$

for observed reflections and

$$\sigma_F' = \frac{1}{(n)^{1/2}} \frac{1}{2(Lp)^{1/2}} (BG)^{1/2}$$

for unobserved reflections, where:

$n$	number of reflections averaged
$Lp$	Lorentz and polarization corrections
$N_T$	total counts measured on the scan through the peak
$BG$	(background counts accumulated in 40 sec) $\times$ (scan time/40).

This produced a weighted  $R$  somewhat lower than the conventional  $R$  from the unweighted refinement, but did not help the convergence problem.

A careful examination of the model revealed that it had become nearly centrosymmetric. A statistical analysis of the magnitudes of the normalized observed structure factors indicated that a true center of symmetry was present. Accordingly, the atomic  $z$ -coordinates were altered so as to place the center at the origin and the probable space group became  $P6_3/mcm$  (No. 193). There was no further difficulty in obtaining convergence.

Weighted refinement, however, still resulted in nonpositive definite thermal parameters for one niobium. Hence, the final refinement was done with unit weights. Neutral atom scattering factors with dispersion corrections included, from (4), were used throughout.

The formula  $Ba_6Nb_{14}Si_4O_{50}$  corresponds to a cell in which one oxygen occupies a position logically more suited to a barium; it is surrounded by 15 oxygens, with no niobium or silicon within bonding range. A corresponding site in  $Ba_3Nb_6Si_4O_{26}$  is unoccupied by any species.

TABLE I

ATOMIC PARAMETERS FOR  $Ba_{6+x}Nb_{14}Si_4O_{47}$  SPACE GROUP  $P6_3/mcm$  (NO. 193)<sup>a</sup>

Atom: Position:	Ba(1) 12k	Ba(2) <sup>b</sup> 6g	Nb(1) 12k	Nb(2) 12k	Nb(3) 4d	Si(1) 8h	
$x$	0.5932(2)	0.6117(39)	0.2383(5)	0.2371(3)	1/3	1/3	
$y$	0	0	0	0	2/3	2/3	
$z$	0.1076(1)	1/4	0.0434(1)	0.1833(1)	0	0.1918(3)	
$U^c$		0.592(711)				0.695(176)	
$U_{11}$	1.103(64)		2.401(162)	0.253(83)	0.427		
$U_{22}$	1.132		0.378	0.273	0.427		
$U_{33}$	1.012(59)		0.472(91)	0.452(76)	1.429(184)		
$U_{12}$	0.566(43)		0.189(72)	0.137(57)	0.213(62)		
$U_{13}$	0.159(57)		0.327(98)	-0.098(76)	0		
$U_{23}$	0		0	0	0		
Atom: Position:	O(1) 12k	O(2) 6g	O(3) 12k	O(4) 12k	O(5) 4c	O(6) 24l	O(7) 24l
$x$	0.2203(26)	0.2860(44)	0.2060(38)	0.1773(24)	1/3	0.1801(20)	0.1729(22)
$y$	0	0	0	0	2/3	0.4898(22)	0.4841(23)
$z$	0.1094(7)	1/4	0.5326(9)	0.6850(7)	1/4	0.1694(4)	0.4562(5)
$U$	1.017(358)	1.384(633)	2.213(572)	0.631(369)	1.501(791)	0.752(254)	0.977(295)

<sup>a</sup> Errors in the last digit are given in parentheses. Where no error is shown, the parameter is not independent. Anisotropic thermal vibration expressed as:  $\exp[-1/4(a^*U_{11}h^2 + b^*U_{22}k^2 + c^*U_{33}l^2 + 2a^*b^*U_{12}hk + 2a^*c^*U_{13}hl + 2b^*c^*U_{23}kl)]$ . Isotropic thermal vibration expressed as:  $\exp[-8\pi^2 U \sin^2 \theta / \lambda^2]$ .

<sup>b</sup> Population parameter, 0.076(11).

<sup>c</sup> All  $U$ 's have been multiplied by 100.

In this case, however, leaving the site empty results in a peak on a  $\Delta F$  map, of approximately the magnitude of a peak produced by omission of one of the normal oxygens. Fully occupying the site with oxygen results in a very large isotropic thermal parameter for the oxygen; allowing both population and thermal parameters to vary results in a low occupancy factor and a negative temperature factor. Placing a barium with a low occupancy factor in the site and refining, however, leads to simultaneous convergence of population and thermal parameters. Accordingly, the probable occupant of the site is a barium of low population parameter, and the compound formula becomes  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ , with  $x \approx 0.23$ . The average niobium oxidation state is thus +4.68, presumably corresponding to a mixture of  $\text{Nb}^{\text{IV}}$  and  $\text{Nb}^{\text{V}}$ . Mixed oxidation states are known for other niobium compounds (5), and in this case account neatly for the dark color of the compound. The theoretical density is  $5.09 \text{ g/cm}^3$ , compared to the measured (presumably low) value of  $4.9 \text{ g/cm}^3$ . A final difference map showed no peak of more than 1/10 the height of that produced by omission of an oxygen.

The final  $R$  factor is 0.062 overall, and 0.045 for reflections from the third data collection (presumably the more accurate reflections). The maximum shift/error ratio in the final cycle of refinement was 0.011. The final parameters are given in Table I, and the unit cell contents are shown in stereo in Fig. 1. This and subsequent

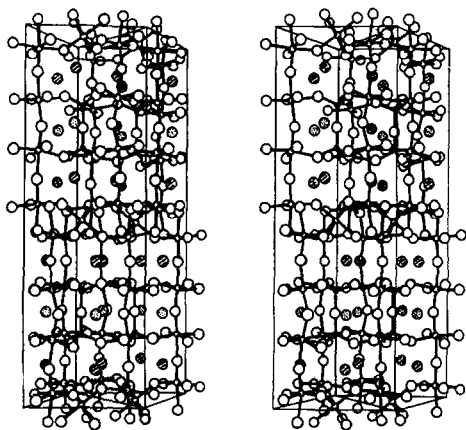


FIG. 1. Stereo view of the unit cell of  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ . Ba(1) is shown as striped circles, Ba(2) as stippled circles, Nb as small open circles, Si as small black circles, and O as large open circles.

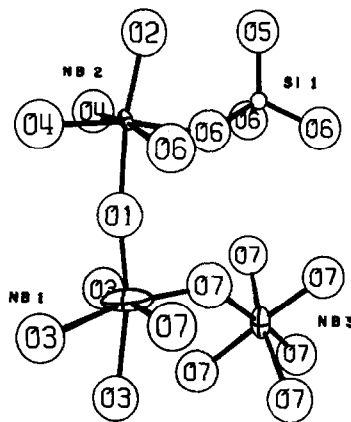


FIG. 2. Niobium and silicon coordination in  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ .

drawings were produced by the program ORTEP (6), using a computer-controlled plotter.

### Discussion

The basic structural unit in  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$  is a string of four  $\text{NbO}_6$  octahedra joined end to end. These strings are joined laterally in groups of three by further corner sharing. Figure 5 shows four such groups and the connections among them. It may be seen that each group of three strings is joined to the next group along the  $c$  axis by edge sharing of the end octahedra. Thus there are essentially continuous triple chains of  $\text{NbO}_6$  octahedra extending along the  $c$  axis. Connections perpendicular to  $c$  between different triple chains are provided by  $\text{Si}_2\text{O}_7$  groups and single  $\text{NbO}_6$  octahedra. Four of the former and two of the latter appear in Fig. 5. Each  $\text{Si}_2\text{O}_7$  group joins three separate triple chains by sharing corners with the second and third octahedra of one string in each group. The single  $\text{NbO}_6$  octahedra occur at the ends of the triple chains, so that each actually shares corners with octahedra in six different four-octahedron strings. This is clearly shown in Fig. 3, which shows a view along the  $c$ -axis of a slice of the unit cell centered around  $z = \frac{1}{2}$ , a region in which edge sharing occurs. The end octahedra of triple chains appear around the origins; single  $\text{NbO}_6$  octahedra are located at  $1/3, 2/3$  and  $2/3, 1/3$ .

In the  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$  structure (1), similar triple chains of niobium-oxygen octahedra appear, but in that case there is no edge sharing; individual chains are infinite in length instead of

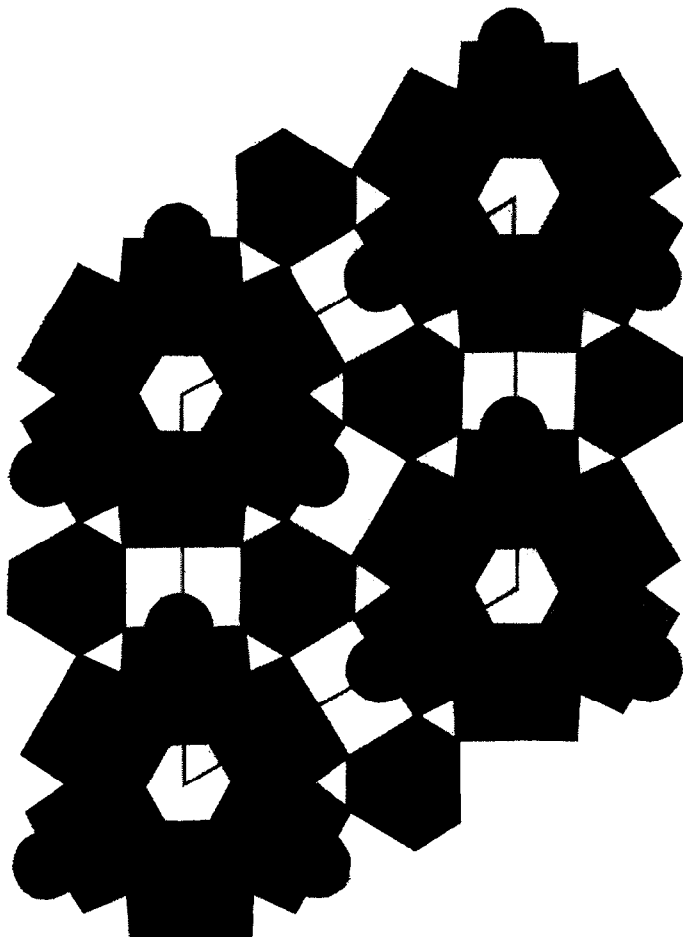


FIG. 3. View along  $c$  axis of a section from  $z = 0.45$  to  $z = 0.60$  of the  $Ba_{6+x}Nb_{14}Si_4O_{47}$  cell, showing  $NbO_6$  groups as solid octahedra and Ba as striped circles.

four octahedra long. The triple chains are joined by linear  $Si_2O_7$  groups, as in the present compound, but there is no niobium between chains.

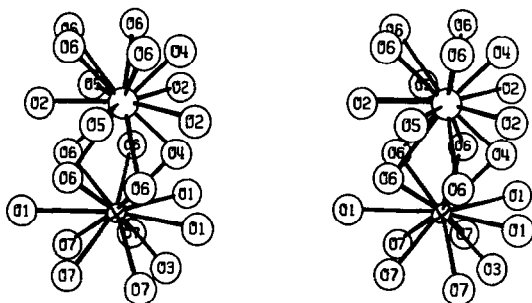


FIG. 4. Stereo view of the barium coordination in  $Ba_{6+x}Nb_{14}Si_4O_{47}$ . Ba(1) is shown as an ellipsoid and Ba(2) as a circle.

The unit cell of  $Ba_3Nb_6Si_4O_{26}$  is two octahedra high; the  $Ba_{6+x}Nb_{14}Si_4O_{47}$  cell may be constructed by stacking up four  $Ba_3Nb_6Si_4O_{26}$  cells, with the top two rotated  $60^\circ$  relative to the bottom two. The second and third cells are then overlapped so that edge sharing of  $NbO_6$  octahedra is produced, the pairs of silicons brought together by this process are replaced by single niobiums, and Ba(2), which occupies a site vacant in  $Ba_3Nb_6Si_4O_{26}$ , is introduced to complete the structure.

So  $Ba_3Nb_6Si_4O_{26}$  is a fairly close relative to  $Ba_{6+x}Nb_{14}Si_4O_{47}$ . It may also be instructive to consider some more distantly related compounds. Except for the  $Si_2O_7$  groups, the framework of the  $Ba_{6+x}Nb_{14}Si_4O_{47}$  structure is composed of  $NbO_6$  octahedra sharing corners, and in some places edges. The occurrence of a framework of

TABLE II

INTERATOMIC DISTANCES AND ANGLES IN  $Ba_{6+x}Nb_{14}Si_4O_{47}^a$ 

## Distances (Å)

Ba(1)–O(1) <sup>1</sup>	3.37(2)
Ba(1)–O(1) <sup>2</sup>	3.19(1)
Ba(1)–O(3)	2.76(3)
Ba(1)–O(4)	2.99(2)
Ba(1)–O(6) <sup>2</sup>	3.04(1)
Ba(1)–O(6) <sup>3</sup>	2.83(2)
Ba(1)–O(7) <sup>2</sup>	3.09(1)
Ba(1)–O(7) <sup>3</sup>	2.84(2)
Ba(2)–O(2) <sup>1</sup>	2.94(5)
Ba(2)–O(2) <sup>2</sup>	3.15(3)
Ba(2)–O(4)	2.63(3)
Ba(2)–O(5)	2.80(1)
Ba(2)–O(6) <sup>2</sup>	3.34(1)
Ba(2)–O(6) <sup>3</sup>	3.27(3)
Nb(1)–O(1)	1.84(2)
Nb(1)–O(3) (atoms at different <i>z</i> values)	2.13(2)
Nb(1)–O(3) (atoms at about same <i>z</i> )	2.04(4)
Nb(1)–O(7)	1.98(2)
Nb(2)–O(1)	2.06(2)
Nb(2)–O(2)	1.91(1)
Nb(2)–O(4)	1.93(3)
Nb(2)–O(6)	2.07(2)
Nb(3)–O(7)	1.98(2)
Si(1)–O(5)	1.62(1)
Si(1)–O(6)	1.63(2)

## Angles (°)

O(1)–Nb(1)–O(3) [O(3) and Nb at about same <i>z</i> ]	95.4(8)
O(1)–Nb(1)–O(3) [O(3) and Nb at different <i>z</i> 's]	167.1(12)
O(1)–Nb(1)–O(7)	93.3(7)
O(3)–Nb(1)–O(3) (O's at same <i>z</i> )	104.0(12)
O(3)–Nb(1)–O(3) (O's at different <i>z</i> 's)	76.9(9)
O(3)–Nb(1)–O(7) [O(3) and Nb at about same <i>z</i> ]	84.1(10), 167.5(10)
O(3)–Nb(1)–O(7) [O(3) and Nb at different <i>z</i> 's]	96.1(8)
O(7)–Nb(1)–O(7)	86.5(7)
O(1)–Nb(2)–O(2)	170.8(13)
O(1)–Nb(2)–O(4)	88.4(7)
O(1)–Nb(2)–O(6)	82.3(6)
O(2)–Nb(2)–O(4)	97.9(10)
O(2)–Nb(2)–O(6)	91.0(9)
O(4)–Nb(2)–O(4)	91.9(8)
O(4)–Nb(2)–O(6)	90.5(8), 170.4(7)
O(6)–Nb(2)–O(6)	85.7(6)
O(7)–Nb(3)–O(7) (O's at same <i>z</i> )	86.0(7)
O(7)–Nb(3)–O(7) [O's belonging to two Nb(1) octahedra which share edges]	87.6(7)
O(7)–Nb(3)–O(7) (O's at different <i>z</i> 's not belonging to edge-sharing octahedra)	101.3(7), 170.0(10)
O(5)–Si(1)–O(6)	112.5(5)
O(6)–Si(1)–O(6)	106.2(9)
Si–O–Si	180

<sup>a</sup> Errors in last digit are shown in parentheses. Superscripts on oxygens around Ba(1) and Ba(2) refer to the order in which the symmetry-related coordinates of the general position for  $P6_3/mcm$  are listed in (11).

corner-sharing  $\text{MO}_6$  octahedra is widespread in oxide structures. Examples include the  $\text{ReO}_3$  structure (7), the hexagonal tungsten bronze structure (8), and the tetragonal tungsten bronze structure (9). The last-named is of particular interest here. Infinite strings of  $\text{WO}_6$  octahedra are joined by a complex pattern of inter-string corner-sharing, so that a projection along the strings shows three-, four- and five-sided rings of octahedra, each ring actually corresponding to an infinite channel bounded by strings of octahedra. Large cations, e.g.,  $\text{K}^+$ , occupy the four-

and five-sided channels, being coordinated by 12 and  $(10 + 5)$  oxygens, respectively.

In  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$  and in the regions of  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$  where edge sharing does not occur, we have nearly identical three-sided channels, as well as five-sided channels that are similar, but have two "sides" composed of  $\text{Si}_2\text{O}_7$  groups instead of octahedra. Because of this substitution, Ba(1) of  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$  and the barium in  $\text{Ba}_3\text{Nb}_6\text{Si}_4\text{O}_{26}$  are coordinated by only  $(10 + 3)$  oxygens, rather than having the  $(10 + 5)$  coordination of Ba(2) and the bronze  $\text{K}^+$ .

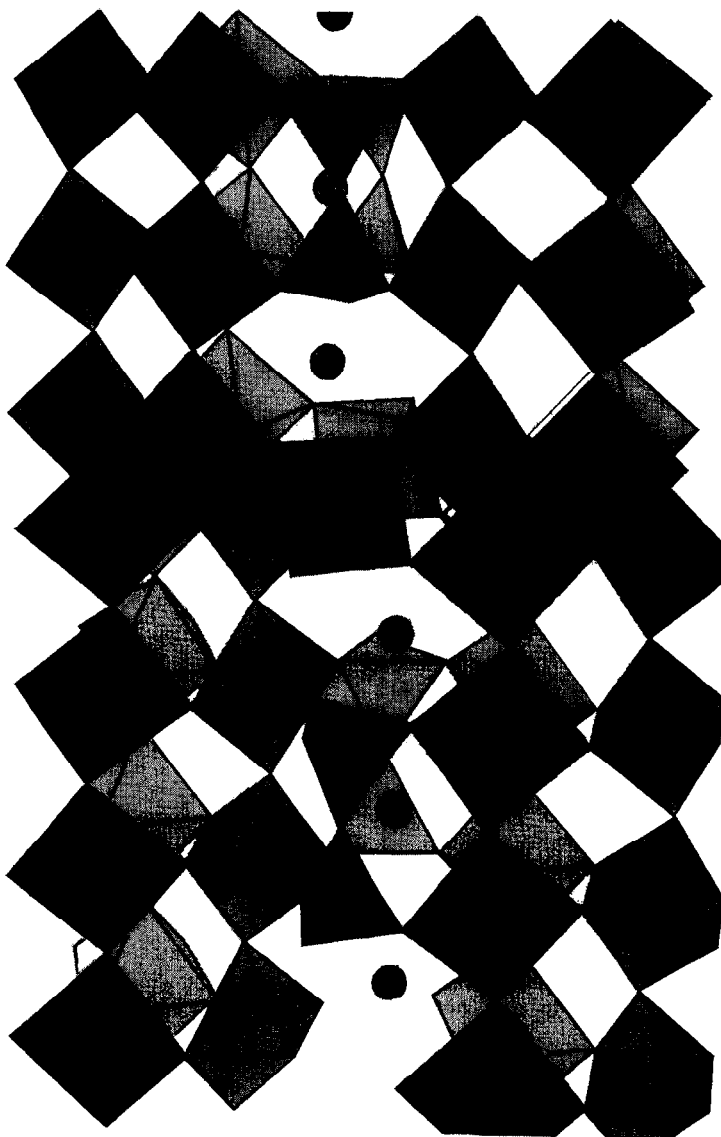


FIG. 5. Arrangement of  $\text{NbO}_6$  octahedra and  $\text{SiO}_4$  tetrahedra in  $\text{Ba}_{6+x}\text{Nb}_{14}\text{Si}_4\text{O}_{47}$ . Bariums are represented as in Fig. 1.

TABLE III

Observed AND Calculated Structure Factors for Ba<sub>6+3</sub>Nb<sub>14</sub>Si<sub>4</sub>O<sub>47</sub><sup>a</sup>

00•L	0•5•L	5 1991 -2042	1•4•L	8 512* -210	10 1263 1286	17 1507 1538	3•4•L	25 1274 -1022	8 1937 -2039
2 3550 2990	0 6790 7129	6 7424 7823	0 4811 4608	10 754 627	12 537* -295	19 357* -153	0 480 -980	29 1197 1421	3 1081 -922
4 1825 -1876	2 667 825	7 7424 -7901	11 465* -322	11 1216 -1196	13 766 661	20 809* -194	2 813 399	35 1139 -473	22 1599 1070
6 2727 2475	4 3703 -3700	9 428* -670	2 690 674	12 729* 677	14 972 -918	22 1312 -1234	2 513* -142	36 1198 -1489	
8 3395 -3361	6 1996 -1881	10 197* 74	3 612* -600	13 861 -823	15 2957 -2901	25 1037 -1113	3 1136 1203		
10 3485 3289	8 478* 199	11 2956 2830	4 2273 2228	14 641* 424	16 1452 -1475	28 1209 -1126	4 642* 386		
12 2338 -2461	10 2669 2784	12 2267 2100	5 477* -689	15 1699 1675	17 566* -190	29 1674 -1706	5 523* -118	0 1755 1712	5•5•L
14 12452 12356	12 2330 -2390	13 1298 1207	6 3038 2951	16 1364 1254	18 1800 1737	31 1120 -924	6 760 -686	10 1089 993	0 3697 4232
16 2841 2350	14 6783 -6628	14 4252 4052	7 596 520	17 1237 1215	19 976 -932	43 1154 983	7 1672 -1583	16 1160 896	1 803 -966
18 1476 4221	16 492* -179	15 1912 1803	8 2258 -2239	18 335* 3	20 377* 350		8 639* -262		2 843* -789
20 4169 4044	20 1369 1398	16 1315 1303	9 948* -822	19 599* 9	21 1605 -1126		9 587* -256		3 572* -528
22 3584 3372	22 1369 1398	17 1274 -1288	10 607* 32	22 972 983	22 552* -2547		10 89* -813	0 2336 3268	4 1347 -1343
24 1297 -1293	22 1392 -1447	18 1011 -1110	11 585* -170	29 1511 -1330	23 1227 1165	0 1528 -1486	11 1226 1283	1 1096 -1468	6 798* -648
26 3676 3484	24 2752 -2888	19 1865 1825	12 305* -87		24 1156 -1278	2 600* -193	12 469* -362	2 508* 131	7 1624 -1400
28 6011 5823	26 139* 1236	20 475* -78	13 533* -286	1•8•L	25 572* 151	3 691* -428	14 539 -362	3 1562 -1886	8 1770 171
30 1889 -1800	28 3633 3565	21 4406 4285	14 1487 -1338		26 635* -556	4 976 -1010	15 1158 -1104	4 2757 2727	10 2949 2856
32 1929 -2116	30 1331 -1319	22 673 703	15 607* 127	0 591* 307	27 849* 563	5 828* 270	16 615* -693	5 618 704	11 539* -104
34 3463 -3210	32 1729 -1847	23 1124 -974	16 2106 1973	1 1683 1651	28 614* 649	6 1031 -991	17 623 -680	6 3180 3064	12 722* -762
36 2191 2178	34 1508 -1718	24 1467 1519	17 720 715	2 1707 1674	29 1486 1983	7 2811 2829	18 987* 41	7 1714 3102	13 2381 -2345
40 1290 -1267	36 1693 1875	25 2323 -2289	18 533* -461	3 1707 1674	32 1486 -1445	8 911 869	19 537* 96	8 3240 -3124	15 1574 -1305
42 1177 -987	40 1175 -1047	26 550* 208	19 488* -103	4 618 779	36 1090 1121	9 626* 93	20 319* -174	10 770 -505	16 1782 1658
44 2089 2230	42 1663 -1903	27 1202 -1919	20 76* 871	5 708 -726	37 1949 -1233	10 642* 41	21 1068 1077	11 2429 -2493	17 1691 1668
46 1659 1844	52 1173 -1416	28 1508 -1501	21 503* -498	6 1231 1280		11 758* -661	22 523* -467	12 563 533	18 1259 -1307
48 2320 2679		29 1116 -1115	22 3037 2904	7 1468 -1630		12 999* 99	23 587* 308	13 1809 -1950	19 1691 1842
52 1169 -1599	0•6•L	30 1114 -1123	23 566* 178	8 2426 -2414	0 427* -4288	13 749* 749	24 117* 104	14 2033 -2033	20 1351 1474
		31 1140 -1123	24 748 730	9 574* 574	1 94* -962	14 659* 659	25 961 -920	15 1614 1298	21 1931 -1931
		32 1062 1659	25 563* -29	10 1695 -1705	2 576* -544	15 1064 -1033	29 236* 1025	16 1078 -944	22 2206 2176
		33 1198 -1095	26 1650 1586	11 1774 1863	3 1816 -1869	16 930 -1002		17 1841 1298	23 1040 -1072
		4 1882 1940	27 602* 195	12 600* -399	4 618* -81		3•5•L	18 981 1171	24 1102 1102
		6 898* -299	28 82* 762	15 1409 -1475	5 103* 103		18 1201 -1249	19 1691 1776	25 1351 1309
		8 1613 1319		1•5•L	6 318* -81		2•8•L	20 2562 -2470	26 2452 2452
		30 1417 1462	0 2612 -2634	17 1216 -1332	7 313* 313			21 2222 -2179	27 520* -565
		41 1549 1591	22 539 539	18 1216 -1332	8 635* -186			22 2222 -2179	28 1133 -1002
			1 1827 -1766	19 1585 -1963	9 565* 106			23 511* 116	29 1949 -1964
			2 598* -657	27 1166 -917	10 618* -1696			24 1199 954	30 1431 1436
			3 523* -1417	29 1084 1065	11 2071 -2015			25 2021 2481	31 1315 1456
			4 122* 208	30 1476 1476	12 3061 -3116			26 1302 1149	32 1554 1483
			5 872 -886	39 1204 1141	13 1371 -1255			27 1312 1495	33 1691 1766
			6 370* -106	40 570* -106	14 2650 2509			28 1410 -1495	34 1335 1309
			4 841 -798		15 837 674			35 1363 -1131	
			5 423* -386	8 511* -292	16 949 -906			36 1255 1227	
			6 1219 -1203	9 541* -106	10 116* -846			37 1402 -1421	
			7 400* -319	10 909 -808	11 1033 -870			38 1351 -1724	
			8 1034 1023	11 1034 862	12 1324 1136			39 1281 1241	
			9 1056 959	12 579* -115	29 1280 -960			40 1216 -1023	
			10 1300 -1231	13 632* -282				41 1201 -1023	
			11 2452 -2497	14 1234 1159	1•10•L			42 1212 -1023	
			12 598 530	15 2322 2187				43 1212 -1023	
			13 1678 -1734	16 463* -950	1 1138 -1135			44 1212 -1023	
			14 2309 2176	17 570* 172	4 1103 1111			45 1212 -1023	
			15 959 -1088	18 1024 -1088	5 2162 -2667			46 1212 -1023	
			17 488 880	19 488 880	6 1416 -1436			47 1212 -1023	
			20 1130 -1061	20 1130 -1061	7 2162 -2667			48 1212 -1023	
			21 1007 988	21 1007 988	8 976* -358			49 1212 -1023	
			22 1137 1137	22 1137 1137	9 872* -903			50 1212 -1023	
			23 814* 789	23 814* 789	10 872* -903			51 1212 -1023	
			24 864 -861	24 864 -861	11 872* -903			52 1212 -1023	
			25 946* -73	25 946* -73	12 872* -903			53 1212 -1023	
			26 612* -40	26 612* -40	13 872* -903			54 1212 -1023	
			27 719* -902	27 719* -902	14 872* -903			55 1212 -1023	
			28 672* -901	28 672* -901	15 872* -903			56 1212 -1023	
			29 613 -689	29 613 -689	16 872* -903			57 1212 -1023	
			30 1941 -1919	30 1941 -1919	17 872* -903			58 1212 -1023	
			31 591* -1540	31 591* -1540	18 872* -903			59 1212 -1023	
			32 2272 -2277	32 2272 -2277	19 872* -903			60 1212 -1023	
			33 278* -2808	33 278* -2808	20 872* -903			61 1212 -1023	
			4 872 750	4 872 750	21 872* -903			62 1212 -1023	
			5 461* 82	5 461* 82	22 872* -903			63 1212 -1023	
			6 418* -562	6 418* -562	23 872* -903			64 1212 -1023	
			7 471* -168	7 471* -168	24 872* -903			65 1212 -1023	
			8 3056 2934	8 3056 2934	25 872* -903			66 1212 -1023	
			9 461* 82	9 461* 82	26 872* -903			67 1212 -1023	
			10 791* -711	10 791* -711	27 872* -903			68 1212 -1023	
			11 1324 -1256	11 1324 -1256	28 872* -903			69 1212 -1023	
			12 1064 1056	12 1064 1056	29 872* -903			70 1212 -1023	
			4 626* 519	4 626* 519	30 872* -903			71 1212 -1023	
			5 794* -782	5 794* -782	31 872* -903			72 1212 -1023	
			6 4612 -4662	6 4612 -4662	32 872* -903			73 1212 -1023	
			7 638* 325	7 638* 325	33 872* -903			74 1212 -1023	
			8 559* 431	8 559* 431	34 872* -903			75 1212 -1023	
			9 1437 -1277	9 1437 -1277	35 872* -903			76 1212 -1023	
			10 1437 -1277	10 1437 -1277	36 872* -903			77 1212 -1023	
			11 1437 -1277	11 1437 -1277	37 872* -903			78 1212 -1023	
			12 1437 -1277	12 1437 -1277	38 872* -903			79 1212 -1023	
			13 1437 -1277	13 1437 -1277	39 872* -903			80 1212 -1023	
			14 1437 -1277	14 1437 -1277	40 872* -903			81 1212 -1023	
			15 1437 -1277	15 1437 -1277	41 872* -903			82 1212 -1023	
			16 1437 -1277	16 1437 -1277	42 872* -903			83 1212 -1023	
			17 1437 -1277	17 1437 -1277	43 872* -903			84 1212 -1023	
			18 1437 -1277	18 1437 -1277	44 872* -903			85 1212 -1023	
			19 1437 -1277	19 1437 -1277	45 872* -903			86 1212 -1023	
			20 1437 -1277	20 1437 -1277	46 872* -903			87 1212 -1023	
			21 1437 -1277	21 1437 -1277	47 872* -903			88 1212 -1023	
			22 1437 -1277	22 1437 -1277	48 872* -903			89 1212 -1023	
			23 1437 -1277	23 1437 -1277	49 872* -903			90 1212 -1023	
			24 1437 -1277	24 1437 -1277	50 872* -903			91 1212 -1023	
			25 1437 -1277	25 1437 -1277	51 872* -903</				



The barium coordination in the present compound is shown in stereo in Fig. 4. Each barium has 10 oxygen neighbors located at the vertices of a pentagonal prism, plus 3 [for Ba(1)] or 5 [for Ba(2)] more oxygens located outside three or all five, respectively, of the side faces of the prism. The coordination is not regular; the Ba–O distances range from 2.6 to 3.4 Å (cf. K–O distances of 2.8 to 3.4 Å in the tetragonal bronze  $K_xWO_3$ ).

$Ba_{6+x}Nb_{14}Si_4O_{47}$  differs from the bronzes, of course, in that edge sharing occurs. We may consider this compound, however, as being constructed from blocks of a bronze-like structure (essentially that of  $Ba_3Nb_6Si_4O_{26}$ ), which are joined by edge sharing, much as blocks of  $ReO_3$ -type structure are joined by edge sharing in such compounds as  $TiNb_2O_7$  and  $TiNb_{24}O_{62}$  (10).

The arrangement of oxygen around niobium and silicon is shown in Fig. 2, and Table II lists interatomic distances and angles. As shown, the  $NbO_6$  octahedra are somewhat distorted. In the case of Nb(1) and Nb(3), this is presumably due to the necessity of fitting so many octahedra together in the edge-sharing region. The distortion, and in particular the tilt away from the vertical, of the Nb(2) octahedra are due primarily to the fact that the oxygens of the  $Si_2O_7$  unit are tightly bound by the silicons, so that the length of the unit is more or less fixed at 4.5 Å. If the octahedra were undistorted and aligned, this would be the Nb(2)–O(2)–Nb(2) distance, corresponding to an Nb–O distance of 2.25 Å. This is rather long compared to the usual Nb–O distance of 1.9–2.1 Å. The tilt of the octahedra brings the niobiums closer together and reduces the Nb(2)–O(2) distance to a reasonable 1.91 Å. In the  $Si_2O_7$  unit, the Si–O distances of 1.62 and 1.63 Å are normal for silicon-containing oxides, and the O–Si–O angles show only slight deviations from the tetrahedral angle of 109.5°. The central Si–O–Si bond is constrained to be linear, as all

three atoms involved are located on a threefold symmetry axis. The linear  $Si_2O_7$  group is rather uncommon, but it has also been found in, for instance,  $Ba_3Nb_6Si_4O_{26}$  (and the corresponding tantalum compound),  $Sc_2Si_2O_7$  (12), and  $NaBa_3Si_2O_7(OH)$  (13), and the evidence is convincing that it does exist in  $Ba_{6+x}Nb_{14}Si_4O_{47}$ .

Table III lists observed and calculated structure factors for the compound.

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